

1970 observed reflections- Coghi *et al.* do not list the atomic parameters, but the over-all structure, including the bond distances and angles, is in reasonable agreement with our results.

Ward *et al.* describe their structure in a unit cell that is slightly more convenient than the one we have chosen ( $a=6.905(5)$ ,  $b=11.680(4)$ ,  $c=17.993(23)$  Å,  $\beta=100.68(10)^\circ$ ; space group,  $P2_1/n$ ).

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## Refinement of the Crystal Structure of Iron Oxychloride

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The lattice constants and the positional parameters of crystals of the compound FeOCl, the crystal structure of which was reported more than 35 years ago, have been redetermined from single-crystal X-ray diffraction data. Except for corrections to the lattice constants and positional parameters, the previous description of the structure remains valid. The most probable space group is *Pmmn*. The revised lattice constants, one of which differs substantially from those previously reported, are  $a=3.780 \pm 0.005$ ,  $b=7.917 \pm 0.005$ , and  $c=3.302 \pm 0.005$  Å. The positional and anisotropic thermal parameters were refined by the method of least-squares, with 294 non-zero diffractometer data, to a conventional  $R=0.055$ . Based on the refined parameters, the  $\text{Fe}^{3+}-\text{O}^{2-}$  bond distances are  $1.964 \pm 0.008$  and  $2.100 \pm 0.010$  Å, and the  $\text{Fe}^{3+}-\text{Cl}^-$  bond distance is  $2.368 \pm 0.007$  Å.

### Introduction

The crystal structure of the compound FeOCl, determined many years ago (Goldsztaub, 1934, 1935), was refined because a more accurate description of the structure was required for a proposed (Muir & Wiedersich, 1967*a*) redetermination of the nuclear quadrupole moment of the 14.4 keV level of  $^{57}\text{Fe}$  ( $Q^{57m}\text{Fe}$ ) from FeOCl data.

This investigation may be expected to aid in resolving the large discrepancies among values of  $Q^{57m}\text{Fe}$  determined from data for other compounds [see discussions by Grant (1966) and by Artman, Muir & Wiedersich (1968)] because FeOCl has certain features which make it especially suitable for determining this constant. An unusually large ferric ion nuclear quadrupole interaction is observed in the  $^{57}\text{Fe}$  Mössbauer spectrum of FeOCl (Muir & Wiedersich, 1967*a*). Furthermore, evaluation

of the electric field gradient tensor at the ferric ion sites in the crystal, which is required in the analysis, is greatly facilitated by the small number of variable positional parameters in the FeOCl crystal structure and by the constraints imposed by the crystal symmetry and ferric ion site symmetry (see below).

In the proposed analysis, the particular step that requires very accurate structural data is a lattice sum calculation of the electric field gradient tensor components. The high sensitivity of such calculations to small variations in structural parameters has been discussed previously (Grant, 1966; Muir & Wiedersich, 1967*b*; Artman *et al.*, 1968).

### Experimental

Crystals of FeOCl were grown by G. P. Espinosa of this Laboratory by a procedure similar to that reported

by Goldsztaub (1935). A mixture of  $\text{FeCl}_3$  and  $\text{Fe}_2\text{O}_3$  in one end of an evacuated and sealed glass tube was maintained at  $350^\circ$ , and the other end of the tube about  $25^\circ$  cooler, for two weeks. The  $\text{FeOCl}$  crystals, which grew in the cooler end of the tube, were similar in their thin blade-like habit and red color to those described by Goldsztaub.

From Buerger precession photographs taken at  $23^\circ\text{C}$ , the most probable space group,  $Pmnm-D_{2h}^{13}$ , reported by Goldsztaub (1934, 1935), was confirmed, and the lattice constants were measured. The crystallographic data are summarized in Table 1. The presently reported values of  $a$  and  $c$  agree satisfactorily with the values, 3.75 and 3.3 Å respectively, given by Goldsztaub (1934, 1935). The presently reported value of  $b$  agrees reasonably well with the value of 7.95 Å in the first of Goldsztaub's two papers on the  $\text{FeOCl}$  structure, but differs substantially from his subsequently revised value, in the second paper, of 7.65 Å, which must be concluded to be erroneous. As Goldsztaub (1935) pointed out, the thinness of the crystals in the  $b$  direction causes difficulty in determining  $b$  from  $b$ -axis rotation photographs. However, there is no such difficulty in the case of the Buerger precession photographs, from which the value of  $b$  appeared to be determinable with about the same accuracy as  $a$  and  $c$ . The presently reported values of the lattice constants were confirmed during the course of diffractometer measurement of the intensities.

Table 1. *Crystallographic data for FeOCl*

Space group:	$Pmnm-D_{2h}^{13}$
Lattice constants:	$a = 3.780 \pm 0.005$ Å
	$b = 7.917 \pm 0.005$
	$c = 3.302 \pm 0.005$
Unit-cell volume =	$98.82$ Å <sup>3</sup>
Unit cell contains	2FeOCl
Formula weight =	107.3
Calculated density =	$3.606$ g.cm <sup>-3</sup>
Measured density =	$3.55$ g.cm <sup>-3</sup> (Goldsztaub, 1935)
Systematic absences:	$h0l$ with $h+l$ odd

Intensities were measured with a Buerger-Supper-Pace-Picker automatic diffractometer, Mo  $K\alpha$  radiation, and balanced Zr and Y filters. The crystal was rotated about the  $c$  axis. The continuous scan mode was used, with a scan rate of  $1^\circ$  per minute, scan interval of  $2^\circ$  or more, and background counts of 60 seconds or more at the beginning and at the end of the scan interval. All independent  $I(hkl)$  with  $(\sin \theta)/\lambda \leq 0.99$ , a total of 500, were measured. Of these, 308 were greater than the background intensity. One reflection, 010, partially cut off by the beam stop, was estimated visually from a precession photograph. Unfortunately, it was found during the course of the refinement that the scan interval had not been made wide enough for the reflections 112, 113, 114, 204, 115, 225, 016, 116, 046, and 206, causing their measured intensities to be lower than their true intensities. It was decided to omit these ten reflections from the refine-

ment calculations, rather than to remeasure them, because the remaining data were more than adequate to solve the problem.

The crystal used for the intensity measurements was a thin rectangular plate with dimensions  $0.42$  (parallel to rotation axis,  $c$ )  $\times 0.10 \times 0.005$  mm. The intensities were corrected for absorption by the method of Burnham (1966). The linear absorption coefficient for Mo  $K\alpha$  radiation is  $86.6$  cm<sup>-1</sup>. The correction factors were between 1.04 and 1.79.

The intensities were reduced to relative  $|F_o(hkl)|^2$  by application of the usual Lorentz-polarization and Tunell factors.

### Refinement of the structural parameters

All the atoms are in twofold  $mm$  special positions of space group  $Pmnm$ : the two  $\text{Fe}^{3+}$  ions are in the positions ( $b$ ) with coordinates  $0, y, \frac{1}{2}$  and  $\frac{1}{2}, \bar{y}, 0$ ; the two  $\text{O}^{2-}$  and two  $\text{Cl}^-$  ions are in two sets of positions ( $a$ ) with coordinates  $0, y, 0$  and  $\frac{1}{2}, \bar{y}, \frac{1}{2}$ . Thus there are three variable positional parameters, one for each of the three atoms comprising the asymmetric unit. There are nine anisotropic thermal parameters, three for each of the three atoms.

The positional parameter values given by Goldsztaub (1934, 1935),  $y_{\text{Fe}} = 0.097$ ,  $y_{\text{O}} = -0.083$ , and  $y_{\text{Cl}} = 0.305$ , were taken as the starting values for the refinement.

The parameters were refined by the method of least-squares with the Busing, Martin & Levy (1962) computer program *ORFLS*. The function minimized was  $\sum w(|F_o| - (1/k)|F_c|)^2$ , in which  $k$  is a scale factor and  $w$  is a weighting function (for  $|F_o| \geq 14$ ,  $w = 200/|F_o|^2$ ; for  $14 > |F_o| \geq |F_{o,\text{min}}|$ ,  $w = 1$ ). Only the  $|F_o|$  above background were included in the least-squares calculations. Fourteen non-zero reflections were omitted from the least-squares calculations: 010, which was estimated visually; 002, 004, and 006, for which peak-heights rather than integrated intensities were measured; and the ten reflections mentioned earlier. However, during all refinement cycles, structure factors were calculated for the unobserved data and for the omitted non-zero reflections.

Scattering factors and dispersion corrections for  $\text{Fe}^{3+}$  and  $\text{Cl}^-$  were taken from *International Tables for X-ray Crystallography* (1962); for  $\text{O}^{2-}$ , the scattering factors given by Tokonami (1965) were used.

The refinement was carried out first with isotropic and finally with anisotropic thermal parameters. The procedure was repeated with atomic, rather than ionic, scattering factors, but this did not produce any significant changes in the parameter values or in the final value of the agreement index  $R = \sum ||F_o| - |F_c|| / \sum |F_o|$ . The final parameter values are given in Table 2. All three positional parameters differ substantially from the previously reported values. Comparison of the  $|F_o|$  with the  $|F_c|$  calculated from the final parameter values is shown in Table 3. The final value of  $R$  is 0.055

for the data included in the least-squares calculation. For the unobserved reflections and the other reflections omitted from the least-squares calculations, Table 3 shows the agreement of  $|F_o|$  and  $|F_c|$  to be satisfactory.

Table 2. Positional and thermal parameters and their estimated standard errors\*

	$10^5 y$	$10^5 \beta_{11} \uparrow$	$10^5 \beta_{22}$	$10^5 \beta_{33}$
Fe <sup>3+</sup>	11568 (13)	791 (35)	362 (11)	2481 (64)
O <sup>2-</sup>	-4832 (61)	1113 (189)	293 (46)	2432 (324)
Cl <sup>-</sup>	33003 (22)	3649 (105)	392 (17)	2403 (119)

\* The estimated standard errors ( $\times 10^5$ ) are given in parentheses.

† The  $\beta_{ii}$  are for the temperature factor expression  $\exp[-(\beta_{11}h^2 + \beta_{22}k^2 + \beta_{33}l^2 + 2\beta_{12}hk + 2\beta_{13}hl + 2\beta_{23}kl)]$ ; for all three ions the site symmetry requires  $\beta_{12} = \beta_{13} = \beta_{23} = 0$ .

Discussion of the structure

The structure is illustrated in Fig. 1. Bond distances, bond angles, and other interatomic distances, computed with the Busing, Martin & Levy (1964) computer program ORFFE, are given in Table 4.

Except for the corrections to the lattice constants and positional parameters found in the present investigation, Goldsztaub's (1934, 1935) description of the structure remains valid. The structure consists of neutral layers, of composition FeOCl, oriented perpendic-

Table 3. Comparison of  $10 \times |F_o|$  with  $10 \times |F_c|$  calculated from final parameter values

Also listed are the calculated values of  $10 \times \alpha = 10 \times \tan^{-1}(B/A)$ . Within each reciprocal lattice level  $l$  the data are in the order of increasing detector angle. The 14 non-zero data omitted from the least-squares calculations are indicated by asterisks.

l = 0		l = 1		l = 2		l = 3		l = 4		l = 5	
$h$	$k  F_o   F_c $	$h$	$k  F_o   F_c $	$h$	$k  F_o   F_c $	$h$	$k  F_o   F_c $	$h$	$k  F_o   F_c $	$h$	$k  F_o   F_c $
0	0	0	0	0	0	0	0	0	0	0	0
1	1340 356 14	1	1340 356 14	1	1340 356 14	1	1340 356 14	1	1340 356 14	1	1340 356 14
2	2680 712 28	2	2680 712 28	2	2680 712 28	2	2680 712 28	2	2680 712 28	2	2680 712 28
3	4020 1068 42	3	4020 1068 42	3	4020 1068 42	3	4020 1068 42	3	4020 1068 42	3	4020 1068 42
4	5360 1424 56	4	5360 1424 56	4	5360 1424 56	4	5360 1424 56	4	5360 1424 56	4	5360 1424 56
5	6700 1780 70	5	6700 1780 70	5	6700 1780 70	5	6700 1780 70	5	6700 1780 70	5	6700 1780 70
6	8040 2136 84	6	8040 2136 84	6	8040 2136 84	6	8040 2136 84	6	8040 2136 84	6	8040 2136 84
7	9380 2492 98	7	9380 2492 98	7	9380 2492 98	7	9380 2492 98	7	9380 2492 98	7	9380 2492 98
8	10720 2848 112	8	10720 2848 112	8	10720 2848 112	8	10720 2848 112	8	10720 2848 112	8	10720 2848 112
9	12060 3204 126	9	12060 3204 126	9	12060 3204 126	9	12060 3204 126	9	12060 3204 126	9	12060 3204 126
10	13400 3560 140	10	13400 3560 140	10	13400 3560 140	10	13400 3560 140	10	13400 3560 140	10	13400 3560 140
11	14740 3916 154	11	14740 3916 154	11	14740 3916 154	11	14740 3916 154	11	14740 3916 154	11	14740 3916 154
12	16080 4272 168	12	16080 4272 168	12	16080 4272 168	12	16080 4272 168	12	16080 4272 168	12	16080 4272 168
13	17420 4628 182	13	17420 4628 182	13	17420 4628 182	13	17420 4628 182	13	17420 4628 182	13	17420 4628 182
14	18760 4984 196	14	18760 4984 196	14	18760 4984 196	14	18760 4984 196	14	18760 4984 196	14	18760 4984 196
15	20100 5340 210	15	20100 5340 210	15	20100 5340 210	15	20100 5340 210	15	20100 5340 210	15	20100 5340 210
16	21440 5696 224	16	21440 5696 224	16	21440 5696 224	16	21440 5696 224	16	21440 5696 224	16	21440 5696 224
17	22780 6052 238	17	22780 6052 238	17	22780 6052 238	17	22780 6052 238	17	22780 6052 238	17	22780 6052 238
18	24120 6408 252	18	24120 6408 252	18	24120 6408 252	18	24120 6408 252	18	24120 6408 252	18	24120 6408 252
19	25460 6764 266	19	25460 6764 266	19	25460 6764 266	19	25460 6764 266	19	25460 6764 266	19	25460 6764 266
20	26800 7120 280	20	26800 7120 280	20	26800 7120 280	20	26800 7120 280	20	26800 7120 280	20	26800 7120 280
21	28140 7476 294	21	28140 7476 294	21	28140 7476 294	21	28140 7476 294	21	28140 7476 294	21	28140 7476 294
22	29480 7832 308	22	29480 7832 308	22	29480 7832 308	22	29480 7832 308	22	29480 7832 308	22	29480 7832 308
23	30820 8188 322	23	30820 8188 322	23	30820 8188 322	23	30820 8188 322	23	30820 8188 322	23	30820 8188 322
24	32160 8544 336	24	32160 8544 336	24	32160 8544 336	24	32160 8544 336	24	32160 8544 336	24	32160 8544 336
25	33500 8900 350	25	33500 8900 350	25	33500 8900 350	25	33500 8900 350	25	33500 8900 350	25	33500 8900 350
26	34840 9256 364	26	34840 9256 364	26	34840 9256 364	26	34840 9256 364	26	34840 9256 364	26	34840 9256 364
27	36180 9612 378	27	36180 9612 378	27	36180 9612 378	27	36180 9612 378	27	36180 9612 378	27	36180 9612 378
28	37520 9968 392	28	37520 9968 392	28	37520 9968 392	28	37520 9968 392	28	37520 9968 392	28	37520 9968 392
29	38860 10324 406	29	38860 10324 406	29	38860 10324 406	29	38860 10324 406	29	38860 10324 406	29	38860 10324 406
30	40200 10680 420	30	40200 10680 420	30	40200 10680 420	30	40200 10680 420	30	40200 10680 420	30	40200 10680 420
31	41540 11036 434	31	41540 11036 434	31	41540 11036 434	31	41540 11036 434	31	41540 11036 434	31	41540 11036 434
32	42880 11392 448	32	42880 11392 448	32	42880 11392 448	32	42880 11392 448	32	42880 11392 448	32	42880 11392 448
33	44220 11748 462	33	44220 11748 462	33	44220 11748 462	33	44220 11748 462	33	44220 11748 462	33	44220 11748 462
34	45560 12104 476	34	45560 12104 476	34	45560 12104 476	34	45560 12104 476	34	45560 12104 476	34	45560 12104 476
35	46900 12460 490	35	46900 12460 490	35	46900 12460 490	35	46900 12460 490	35	46900 12460 490	35	46900 12460 490
36	48240 12816 504	36	48240 12816 504	36	48240 12816 504	36	48240 12816 504	36	48240 12816 504	36	48240 12816 504
37	49580 13172 518	37	49580 13172 518	37	49580 13172 518	37	49580 13172 518	37	49580 13172 518	37	49580 13172 518
38	50920 13528 532	38	50920 13528 532	38	50920 13528 532	38	50920 13528 532	38	50920 13528 532	38	50920 13528 532
39	52260 13884 546	39	52260 13884 546	39	52260 13884 546	39	52260 13884 546	39	52260 13884 546	39	52260 13884 546
40	53600 14240 560	40	53600 14240 560	40	53600 14240 560	40	53600 14240 560	40	53600 14240 560	40	53600 14240 560
41	54940 14596 574	41	54940 14596 574	41	54940 14596 574	41	54940 14596 574	41	54940 14596 574	41	54940 14596 574
42	56280 14952 588	42	56280 14952 588	42	56280 14952 588	42	56280 14952 588	42	56280 14952 588	42	56280 14952 588
43	57620 15308 602	43	57620 15308 602	43	57620 15308 602	43	57620 15308 602	43	57620 15308 602	43	57620 15308 602
44	58960 15664 616	44	58960 15664 616	44	58960 15664 616	44	58960 15664 616	44	58960 15664 616	44	58960 15664 616
45	60300 16020 630	45	60300 16020 630	45	60300 16020 630	45	60300 16020 630	45	60300 16020 630	45	60300 16020 630
46	61640 16376 644	46	61640 16376 644	46	61640 16376 644	46	61640 16376 644	46	61640 16376 644	46	61640 16376 644
47	62980 16732 658	47	62980 16732 658	47	62980 16732 658	47	62980 16732 658	47	62980 16732 658	47	62980 16732 658
48	64320 17088 672	48	64320 17088 672	48	64320 17088 672	48	64320 17088 672	48	64320 17088 672	48	64320 17088 672
49	65660 17444 686	49	65660 17444 686	49	65660 17444 686	49	65660 17444 686	49	65660 17444 686	49	65660 17444 686
50	67000 17800 700	50	67000 17800 700	50	67000 17800 700	50	67000 17800 700	50	67000 17800 700	50	67000 17800 700
51	68340 18156 714	51	68340 18156 714	51	68340 18156 714	51	68340 18156 714	51	68340 18156 714	51	68340 18156 714
52	69680 18512 728	52	69680 18512 728	52	69680 18512 728	52	69680 18512 728	52	69680 18512 728	52	69680 18512 728
53	71020 18868 742	53	71020 18868 742	53	71020 18868 742	53	71020 18868 742	53	71020 18868 742	53	71020 18868 742
54	72360 19224 756	54	72360 19224 756	54	72360 19224 756	54	72360 19224 756	54	72360 19224 756	54	72360 19224 756
55	73700 19580 770	55	73700 19580 770	55	73700 19580 770	55	73700 19580 770	55	73700 19580 770	55	73700 19580 770
56	75040 19936 784	56	75040 19936 784	56	75040 19936 784	56	75040 19936 784	56	75040 19936 784	56	75040 19936 784
57	76380 20292 798	57	76380 20292 798	57	76380 20292 798	57	76380 20292 798	57	76380 20292 798	57	76380 20292 798
58	77720 20648 812	58	77720 20648 812	58	77720 20648 812	58	77720 20648 812	58	77720 20648 812	58	77720 20648 812
59	79060 21004 826	59	79060 21004 826	59	79060 21004 826	59	79060 21004 826	59	79060 21004 826	59	79060 21004 826
60	80400 21360 840	60	80400 21360 840	60	80400 21360 840	60	80400 21360 840	60	80400 21360 840	60	80400 21360 840
61	81740 21716 854	61	81740 21716 854	61	81740 21716 854	61	81740 21716 854	61	81740 21716 854	61	81740 21716 854
62	83080 22072 868	62	83080 22072 868	62	83080						

well for the perfect cleavage plane perpendicular to **b** and for the flexibility of the crystals, which allows them easily to be bent and then returned to their original shape.

Each  $\text{Fe}^{3+}$  ion is bonded to two  $\text{O}^{2-}$  ions at a distance of  $1.964 \pm 0.008 \text{ \AA}$  and to two others at a distance of  $2.100 \pm 0.010 \text{ \AA}$ . The average of these distances is  $2.03 \text{ \AA}$ . These distances are nearly the same as the  $\text{Fe}^{3+}-\text{O}^{2-}$  bond distances in  $\alpha\text{-Fe}_2\text{O}_3$  (hematite), the structure of which was recently refined by Blake, Hessevick, Zoltai & Finger (1966). In  $\alpha\text{-Fe}_2\text{O}_3$  each  $\text{Fe}^{3+}$  ion is bonded to three  $\text{O}^{2-}$  ions at a distance of  $1.945 \text{ \AA}$  and to three others at a distance of  $2.116 \text{ \AA}$ , in an octahedral arrangement. These distances also average to  $2.03 \text{ \AA}$ . The  $\text{Fe}^{3+}-\text{O}^{2-}$  bond distances in  $\text{FeOCl}$  are also nearly the same as the  $\text{Fe}^{3+}-\text{O}^{2-}$  bond distances reported for  $\gamma\text{-FeOOH}$  (lepidocrocite), the structure of which is closely related to that of  $\text{FeOCl}$ . In  $\gamma\text{-FeOOH}$  there are layers similar to those in  $\text{FeOCl}$ , with  $\text{OH}^-$  ions instead of  $\text{Cl}^-$  ions. The  $\text{Fe}^{3+}-\text{O}^{2-}$  bond distances in  $\gamma\text{-FeOOH}$ ,  $1.94$  and  $2.14 \text{ \AA}$ , which average to  $2.04 \text{ \AA}$ , were reported by Goldsztaub (1935) at the same time as his results on  $\text{FeOCl}$ . These average  $\text{Fe}^{3+}-\text{O}^{2-}$  distances are all somewhat longer than the octahedral  $\text{Fe}^{3+}-\text{O}^{2-}$  distances in garnets (Lind &

Geller, 1969, and references cited therein) and in perovskite-like structures (Coppens & Eibschütz, 1965).

The  $\text{Fe}^{3+}-\text{Cl}^-$  bond distance,  $2.368 \pm 0.007 \text{ \AA}$ , is practically the same as the  $\text{Fe}^{3+}-\text{Cl}^-$  bond distance,  $2.39 \text{ \AA}$ , in  $\text{FeCl}_3$  (Wooster, 1932). It is substantially greater than the  $\text{Fe}^{3+}-\text{Cl}^-$  bond distance of  $2.30 \text{ \AA}$  found in the  $\text{trans-}[\text{FeCl}_2(\text{H}_2\text{O})_4]^+$  ion (Lind, 1967). Interference between  $\text{Cl}^-$  ions in *cis* positions is probably the reason for the longer  $\text{Fe}^{3+}-\text{Cl}^-$  distances in  $\text{FeOCl}$  and  $\text{FeCl}_3$ .

The  $\text{FeCl}_2\text{O}_4$  octahedron in  $\text{FeOCl}$  is considerably distorted. The edge lengths vary from  $2.624$  to  $3.357 \text{ \AA}$ . (The edge lengths not given in Fig. 1 are those of the  $\text{Cl}^--\text{Cl}^-$  and  $\text{O}^{2-}-\text{O}^{2-}$  edges parallel to **c**, which are equal to the lattice constant *c*.) The shortest  $\text{O}^{2-}-\text{O}^{2-}$  and  $\text{O}^{2-}-\text{Cl}^-$  edges of the octahedron,  $2.624 \text{ \AA}$  and  $2.995 \text{ \AA}$  respectively, are those which are shared with other octahedra. The packing of the four  $\text{O}^{2-}$  and two  $\text{Cl}^-$  ions about the  $\text{Fe}^{3+}$  ion is such that the centers of all the  $\text{O}^{2-}$  ions are well within one hemisphere about the  $\text{Fe}^{3+}$  ion, opposite to the  $\text{Cl}^-$  ions. The  $\text{O}^{2-}-\text{Fe}^{3+}-\text{O}^{2-}$  and  $\text{O}^{2-}-\text{Fe}^{3+}-\text{Cl}^-$  angles between bonds directed toward opposite corners of the octahedron are  $148.48$  and  $172.39^\circ$  respectively. These may be compared with an  $\text{O}^{2-}-\text{Fe}^{3+}-\text{O}^{2-}$  angle of  $162.21^\circ$  in the  $\alpha\text{-Fe}_2\text{O}_3$

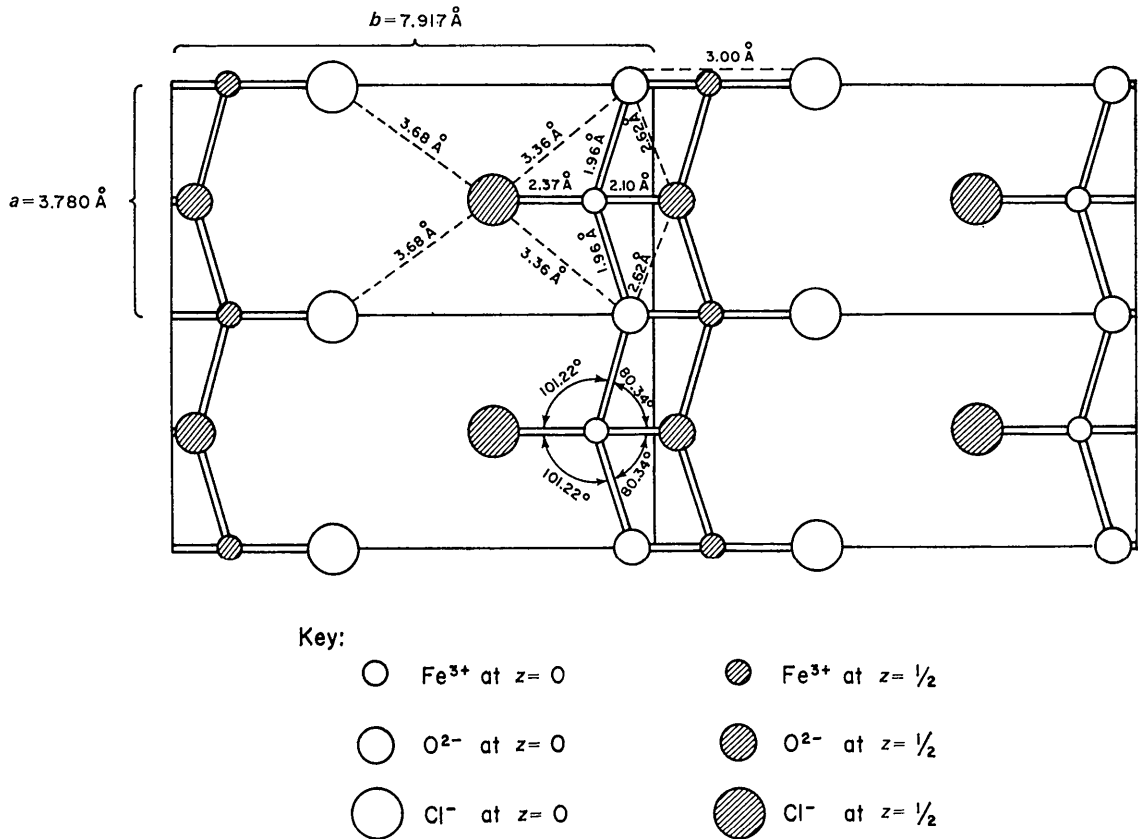


Fig. 1. Projection of structure on (001) showing four complete unit cells.

structure (Blake, *et al.*, 1966). The other bond angles at the Fe<sup>3+</sup> ion are in the range 80.34 to 103.64°, which is about the same as that reported for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (Blake, *et al.*, 1966).

The interlayer Cl<sup>-</sup>-Cl<sup>-</sup> distance, 3.680 Å, closely approximates twice the van der Waals radius of chlorine.

The principal axes of the thermal motion probability ellipsoids are required by the crystal symmetry to be parallel to the crystallographic axes (as are the principal axes of the electric field gradient tensors, which were discussed in the *Introduction*). The root-mean-square components of thermal displacement along the three principal axes of the thermal motion probability ellipsoids are given in Table 5; these were computed with the Busing, Martin & Levy (1964) computer program ORFFE.

Table 5. *Thermal motion probability ellipsoids*

Root-mean-square component of thermal displacement\* along principal axis parallel to

	a	b	c
Fe <sup>3+</sup>	0.076 ± 0.005 Å	0.107 ± 0.005 Å	0.117 ± 0.005 Å
O <sup>2-</sup>	0.090 ± 0.023	0.097 ± 0.023	0.116 ± 0.023
Cl <sup>-</sup>	0.163 ± 0.007	0.112 ± 0.007	0.115 ± 0.008

\* The limits of error are 3 × the estimated standard errors computed by the program ORFFE.

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## The Relation Between $\gamma$ - and $\alpha$ -Chymotrypsin. II.\* Direct Comparison of the Electron Densities at 5.5 Å Resolution

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Extensive comparisons have been made between 5.5 Å resolution electron density maps of the two molecules in the asymmetric unit of  $\alpha$ -chymotrypsin, and between the  $\alpha$ -chymotrypsin electron density and a 5.5 Å electron density map of  $\gamma$ -chymotrypsin. The transformations describing the relation between the respective pairs of molecules have been refined by a least-squares method. On the basis of these comparisons the respective electron density maps are shown to be identical, within experimental error. It is concluded that at this resolution there is no significant evidence of any conformational difference between the two  $\alpha$ -chymotrypsin molecules or between  $\alpha$ - and  $\gamma$ -chymotrypsin.

### Introduction

$\alpha$ -Chymotrypsin ( $\alpha$ CHT) and  $\gamma$ -chymotrypsin ( $\gamma$ CHT) are the end products of the activation of chymotryp-

sinogen A and are distinguished by their respective crystal forms (Kunitz, 1938; Desnuelle, 1960).  $\alpha$ CHT crystallizes at pH 4.4 in a monoclinic space group with two molecules per asymmetric unit and the structure has been determined to atomic resolution (Matthews, Sigler, Henderson & Blow, 1967; Sigler, Blow, Matthews & Henderson, 1968).  $\gamma$ CHT crystallizes at pH 5.6 in a tetragonal space group and a 5.5 Å resolution elec-

\* The first paper of this series is Matthews, Cohen, Silverton, Braxton & Davies (1968).

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